

DAMAGE CONCEPT IN WELDMENTS

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Two scientific methods today, on the problems of deterioration and fracture of materials have been applied: linear elastic fracture mechanics which starts by Griffiths /1/ considering brittle fracture, and continuous damage mechanics, originated from Kachanov's /2/ analysing creep fracture problem. High idealized assumptions both of mentioned approach tried to eliminate Janson and Hult /3/ and farther Janson /4/,/5/ by himself. They proposed a combined approach with intention to describe the interaction between a macroscopic crack and microscopic damage in the homogenous materials. In this paper a possibility of application of these procedure to the weldments problems has been analysed.

INTRODUCTION

A primary problem existing in the field of weldments is expressive inhomogeneity in the material at the zone of welded joint. This inhomogeneity has been expressed as in the structure of material as in important change of mechanical properties, often at the very small space. However, this inhomogeneity has been mainly expressed to the change of the yield stress in each zone related to the base metal.

As analysis of single cracks, located in whichever zone of welded joint, understands the appearance of the plastic zones at the crack tip, it will be a main difficulty to determine this zone.

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STATEMENT OF THE PROBLEM

A thin sheet with butt weldment, transversal to the strain direction, loaded in mode I, has been treated using the modified Dugdale /6/ crack model according to Jansons /5/ procedure for time-independent damage formation. Three independent and substantially different cases are determined: /a/ the crack in the weld metal, /b/ crack in the heat-affected zone, and /c/ crack in base metal. However, in this work, like most interesting, it will be considered a case when the crack just in the fusion line has been assumed /Fig. 1). To simplify the analysis, we assumed that there was no the residual stresses from welding. A plastic zone, which is forming at the crack tip, will have in this case irregular shape by reason of mentioned structural and mechanical anisotropy /before all by reason of different yield strengths/.

We will assume the shape of plastic zone at the crack tip like in Figure 2.

Our further consideration has been considered on the possibility of determination a mechanical properties of materials at whichever point in welded joint /in accordance with Soet and Denys /7/ papers/. The following relations introduced in this case are:

$$\text{Damage: } \omega = (A - A_{ef})/A \quad (1)$$

$$\text{Stress: } \sigma = P/A \quad (2)$$

$$\text{Net stress: } s = P/A_{ef} \quad (3)$$

$$\text{and consequently: } \sigma = s(1 - \omega) \quad (4)$$

Elasto-plastic relation, net stress-strain is assumed independently for weld metal and heat-affected zone /see Fig. 3/.

Relation damage-net stress, is assumed like in the paper /5/. However, by reason of mentioned inhomogeneity, typical for welding procedure, a new greatness, called initial or technological damage, has been introduced. This initial /technological/ damage exist independently from the external load, and is different for each of the three mentioned cases. It depends on the materials nature /basic or additional/, welding procedure, thermal treatment etc. That means that the initial /technological/ damage is the parameter of material. We can assume that it is constant for the assigned welding parameters.

If we denote new greatness with ω_i , we can express it like relation:

$$\omega_i = \epsilon_m / \epsilon_f \quad (6)$$

where is: ϵ_m -max. strain, ϵ_f -strain at fracture in one plane parallel to the weld axis /in our case this is the fusion line/. It has to be mentioned here, that both value ϵ_m and ϵ_f are variable along the direction perpendicular to the weld axis /y/, as is described in mentioned paper /7/, on the series of small specimens which have been cut out successively to the direction of weld axis. In our case, for $y=0$, $\omega_i^2 = \omega_i$. In fact, this relation express a measure of ductility in the materials, which is a very important property when the welded joints under fracture condition has been considered. Obviously, as a value ω_i is nearer to the unit a fracture conditions become more rigorous.

Now, we shall introduce a modified Jansons relation without a plastic zone:

$$\omega' = C_o'(s')^{1/\omega_i'} + \omega_i' \quad (7.1)$$

$$\omega = C_o(s)^{1/\omega_i} + \omega_i \quad (7.2)$$

At the zone boundary: $\omega' = C_o'(s_y') + \omega_i' = \omega_o' + \omega_i'$ (8.1)

$$\omega = C_o(s_y) + \omega_i = \omega_o + \omega_i \quad (8.2)$$

Inside the zone damage is assumed to depend linearly on the crack opening whit a presence of the initial damage:

$$\omega'(x) = \omega_o' + \omega_i' + k'\eta'(x) \quad (9.1)$$

$$\omega(x) = \omega_o + \omega_i + k\eta(x) \quad (9.2)$$

where are ω_o' , ω_o , k , k' and ω_i , ω_i' material parameters.

Introducing relations /9.1/ and /9.2/ in expression /4/ we have:

$$\sigma'(x) = s_y' [1 - \omega_o' - \omega_i' - k'\eta'(x)] \quad (10.1)$$

$$\sigma(x) = s_y [1 - \omega_o - \omega_i - k\eta(x)] \quad (10.2)$$

We shall consider now resulting stress distribution in the sheet due to external load σ_∞ and the stress σ_x acting on the outer flanks of a crack /Fig. 4/ separating our consideration on two parts: $a < x < b$ and $b < x < c$.

Westergards /8/ complex stress functions are used here due to procedure identical like in paper /5// which will not be represented by reason of shortness/. Using two independent integration procedure of stress function we get two relations /whit integration boundaries a-b and b-c/:

$$a/b = \cos [(\bar{u}/2)(\sigma_\infty/s_y')/\gamma^2] \quad (11.1)$$

$$b/c = \cos [(\bar{u}/2)(\sigma_\infty/s_y)/\gamma^2] \quad (11.2)$$

From this two relations we can calculate the plastic zone length b and c , when we first calculate the crack half-opening $\eta'(x)$ and $\eta(x)$ by using Westergards solution for displacement at y -direction

$$\eta(z) = [z \operatorname{Im} \bar{Z} - y(1+\nu) \operatorname{Re} Z] / E \quad (12)$$

with next conditions: $z=x, y=0$.

We shall consider now a possible crack instability criterion. Using equations /9.1/ and /9.2/ we get a values $\omega'(a)$ and $\omega(a)$.

As criterion for crack propagation Janson suggested the next condition: $\omega(a) = 1$.

In our case, this condition is transformed by two in the shape:

$$\omega'(a) = \omega_0' + k' \eta'(a) = 1 - \omega_i' \quad (13.1)$$

$$\omega(a) = \omega_0 + k \eta(a) = 1 - \omega_i \quad (13.2)$$

where is $\omega_i' = \omega_i$. From two values got by this way, the smaller will be valid. However, a particular situation will originate when ω_i is very near /or equal/ to the unit, i.e. when a materials ductility is reduced at zero practically, and no damage in the material provoked by loading. It can be represented that the use of classical Dugdalss approach in this case is justified.

CONCLUSION

The method proposed here, presents one attempt that a combined approach of fracture mechanics and continuous damage mechanics can be applied on the weldments problems.

Introducing a new parametar of material / ω_i / and separations of integration procedure on two independent parts, would make possible that the problems of local anisotropy and lamination in the welded joints be practically overrun.

SYMBOLS USED

ω	=	damage
A	=	surface
σ	=	stress
S	=	net stress
P	=	loading
ϵ	=	strain
C_0, v_0, k	=	parameters of material
n	=	crack half-opening
z	=	complex variable
Z	=	Function of z, derived from stress function
\bar{Z}	=	first integral of Z
I_m	=	imaginary part of complex function
Re	=	real part of complex function
E	=	Young's modulus
ν	=	Poisson's ratio

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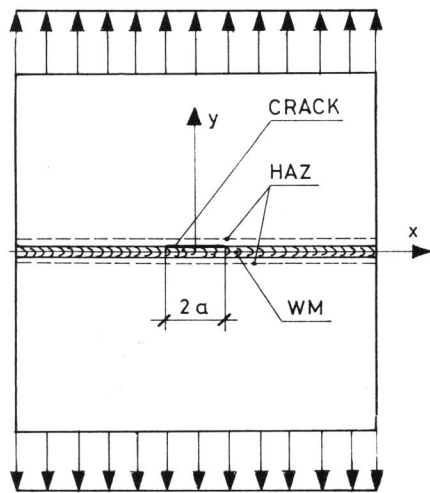


Figure 1

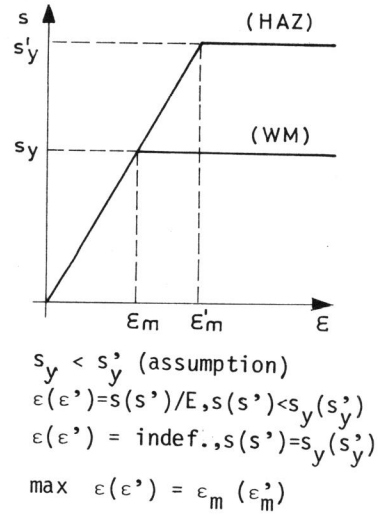


Figure 2

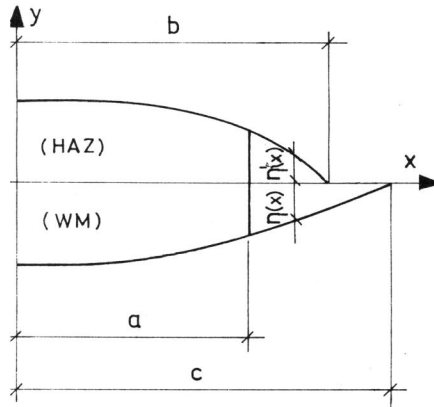


Figure 3

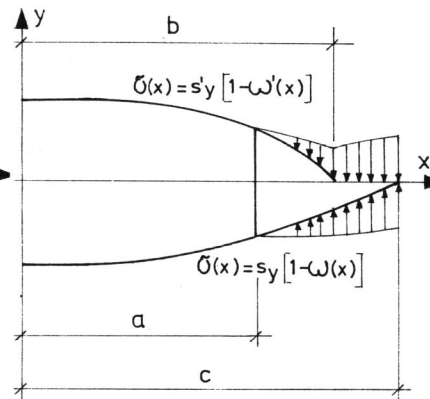


Figure 4