

FRACTURE TOUGHNESS AND RESISTANCE TO FATIGUE CRACK GROWTH OF  
THREE VESSEL STEEL WELDED JOINTS WITH DIFFERENT WELDING PROCEDURES

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INTRODUCTION

Welded-joints of vessel steel plate, made with three different welding procedures and different heat-treatments, were studied using fracture mechanics testing, metallography, electron fractography, etc. The fracture parameters measured are: fracture toughness  $J_{IC}$ , COD,  $\frac{P_{max}}{B(w-a)}$ ,  $da/dN$ , etc. The choice of the three welding procedures and relevant heat-treatment has been made by evaluating and comparing the experimental results.

EXPERIMENTAL PROCEDURE

This vessel steel is a fine-grained steel usually used in nuclear power stations. Vessel steel plate with a thickness of 80 mm was used in our programme. Its composition is shown in Table 1, Table 2 gives the details of welding conditions and heat-treatments of the specimens.

There were two kinds of three-point-bend specimen to be used. The dimension of the first was 38×76×370 mm (according to ASTM E399-76), the other was 20×24×100 mm (GB 2038-80). All 79 specimens were precracked in fatigue with  $a/w = 0.5$ . The positions of the precracked notches are shown in Fig. 1.

The fracture toughness tests were performed on a hydraulic testing machine. Force, clip-gage displacement, potential drop, events of acoustic

emission were recorded on two sets of three X-Y recorders.  $J_{Ic}$ ,  $\frac{P_{max}}{B(w-a)}$  were measured from P- $\Delta$  curves, and COD from P-V curves.

Fatigue crack growth tests were conducted under constant load with stress ratio  $R = 0.2$ , crack length  $a$  and number of cycles  $N$  were recorded from  $a/w = 0.3$  to  $0.5$  (according to ASTM E647-78T).

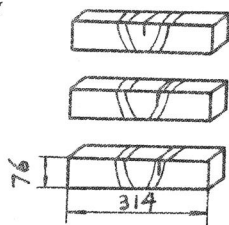


Fig. 1 Positions of notches on specimens

### RESULTS AND DISCUSSION

Fig. 2a and 2b are the results of fracture toughness  $J_{Ic}$ , COD and  $\frac{P_{max}}{B(w-a)}$  of weld metal, fusion line (and its vicinity), HAZ, and parent metal.

The data of fracture toughness of various welded-joints assume a Log-Normal distribution [1]. According to the student t distribution we may evaluate the lower bounds with 99.5% confidence. The results were given in Table 3 [2].

It appears that the weld metal of  $E_1$  is provided with coarse ferrite net structures (see Fig. 3a and b), which result in very poor toughness.

Recent studies revealed that hot strain embrittlement could be one of the reasons for this type of problem in low alloy ferritic steels [3,4]. In our work, this problem may be solved by  $E_2$  heat-treatment provided by the Manufacturing Plant. After this treatment, the welded-joints have fine grains (Fig. 3c) and a little portion of ferrite, which enhance the resistance of crack growth [5] and result in a higher fracture toughness  $J_{Ic}$ .  $J_{Ic}$  of  $E_2$  is about four times higher than that of  $E_1$  in weld metals.

It is noted that the  $J_{Ic}$  of narrow gap welded-joints (as welded) is the same as that of  $S_1$ .

The curves of the fatigue crack growth rate  $da/dN$  versus stress intensity factor range  $\Delta K$  at stage II for the welded joints with different welding procedures are shown in Fig. 4. The fatigue crack growth rates are expressed in the form of Paris formula and some of them are as follows:

$$N_3: \text{ in the vicinity of fusion lines, } \frac{da}{dN} = 0.3491 \times 10^{-10} \Delta K^{3.3272}$$

$$\text{ in H. A. Z. } \frac{da}{dN} = 0.2244 \times 10^{-8} \Delta K^{2.3703}$$

Table 1 Chemical Composition of the Parent and Weld Metals

Composition, %		C	Si	Mn	S	P	Ni	Cr	Mo	Nb	Al	Ti	Cu
Parent Metal		0.15	0.36	1.51	0.003	0.015	0.83	0.35	0.34	0.01	0.031	-	-
Weld Metals	$N_2$	0.1	0.41	1.59	0.014	0.014	0.88	0.094	0.5	0.003	0.01	0.016	0.1
	$N_3$	0.1	0.4	1.6	0.013	0.014	0.89	0.068	0.5	0.003	0.01	0.014	0.1
	$E_1$	0.14	0.26	1.39	0.006	0.018	0.76	0.15	0.31	0.003	0.01	-	V%
	$S_1$	0.05	0.32	1.55	0.016	0.026	0.23	0.062	0.43	0.003	0.01	-	0.26

Table 2 Welding Parameters and Heat-Treatment Conditions

Types of Welding		Current $I_a$ , (A)	Voltage $V$ , (Volt)	Welding Speed $v$ , (m/hr)	Heat Input (KJ/cm)	Type & Diameter of Electrode	Heat-Treatment Conditions
Narrow Gap Welded Joints	$N_1$	430	27	16.5	25.4	H08Mn2SiNiMo, $\phi 3$ mm	*
	$N_2$	"	"	15	27.4		none
	$N_3$	"	"	18	23.22		none
Submerged Arc Welded Joints	$S_1$	bottom 600-650 upper 650-750	28-30 35-38	32 26-28	upper 33.6	H08Mn2MoAl, $\phi 5$ mm	*
	$S_2$	bottom 600-650 upper 650-750	28-30 35-38	32 26-28	"	"	none
Electroslag Welded Joints	$E_1$	500-520	36	1.2	1101.6	H10MnMoNiVA $\phi 3$ mmx2	**
	$E_2$						***

\* Stress relief annealing, 620°Cx8hr, furnace cooling.

\*\* Normalizing, tempering, and stress relief annealing.

\*\*\* Provided by Manufacturing Plant.

Table 3 Mean Values and Standard Deviations of  $\ln J_{Ic}$

Types of Welding	$\bar{\mu}$ of $\ln J_{Ic}$	$\sigma_{n-1}$	Confidence Level	Lower Bounds of $\bar{\mu}$	$J_{Ic}$
N <sub>1</sub>	2.2697	0.1978	99.5%	2.1024	8.186
N <sub>2</sub>	2.3606	0.2060	"	2.1861	8.9
N <sub>3</sub>	2.4259	0.2163	"	2.2599	9.582
S <sub>1</sub>	2.4169	0.2007	"	2.237	9.365
S <sub>2</sub>	2.3314	0.1975	"	2.164	8.706
E <sub>1</sub>	1.816	0.5125	"	1.3264	3.767
E <sub>2</sub>	2.4686	0.128	"	2.433	11.393

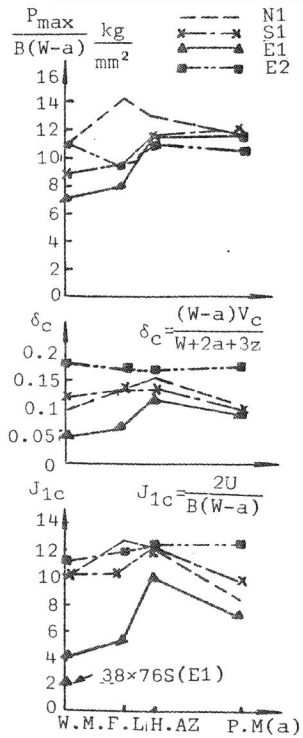


Fig. 2 Comparison of values from different specimens

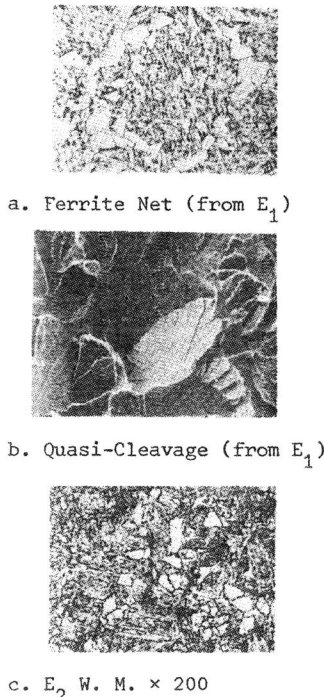


Fig. 3 Structures of weld metals

S<sub>1</sub>: welded metal stress relieved

$$\frac{da}{dN} = 0.88508 \times 10^{-14} \Delta K^{5.3145}$$

vicinity of fusion line

$$\frac{da}{dN} = 0.78453 \times 10^{-13} \Delta K^{4.9641}$$

$$\text{H.A.Z } \frac{da}{dN} = 0.64589 \times 10^{-11} \Delta K^{3.6855}$$

E<sub>2</sub>: vicinity of fusion line

$$\frac{da}{dN} = 0.98273 \times 10^{-9} \Delta K^{2.4463}$$

$$\text{H.A.Z } \frac{da}{dN} = 0.8589 \times 10^{-10} \Delta K^{2.9977}$$

parent metal

$$\frac{da}{dN} = 0.8969 \times 10^{-10} \Delta K^{3.0525}$$

where  $da/dN$  is in mm/cycle,  $\Delta K$  in  $\text{kg} \cdot \text{mm}^{-3/2}$ .

N<sub>3</sub> and S<sub>1</sub> have the same crack growth rate  $da/dN$ . In N.H.T. N<sub>3</sub>, cracks are easy to grow through the boundaries of column crystal, as shown in Fig. 5.

In E<sub>2</sub>, the cracks grow slowly and make a detour from ferrite or stop before ferrite (Fig. 6).

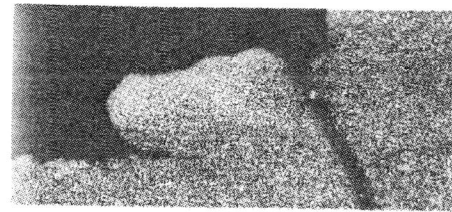


Fig. 5 Cracks grow through the boundaries of column crystal

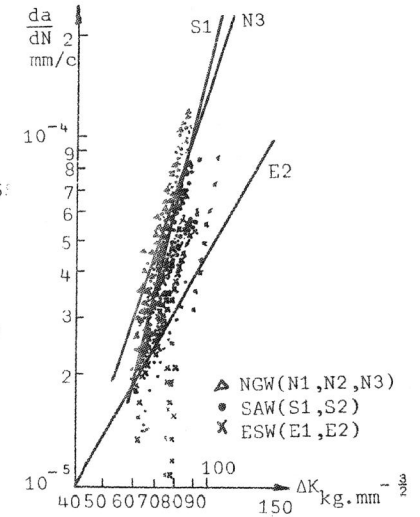


Fig. 4 Fatigue crack growing rates

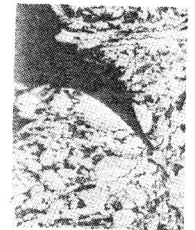


Fig. 6 Crack stops at ferrite crystals

### CONCLUSIONS

1. The weld metal for electroslag welded joints with nor. +temp. +ann. treatment exhibits embrittlement resulting in very poor toughness.
2. This problem of embrittlement may be solved by E<sub>2</sub> heat-treatment of the Manufacturing Plant.

3. The  $J_{IC}$  and  $da/dN$  of narrow gap welded-joints without stress relief have the same values as those of submerged arc welded-joints with stress relief.

4. A little content of ferrite can improve the resistance to fatigue propagating of cracks in electroslog welded-joints.

5. The cracks are easy to grow through the crystal boundaries in narrow gap welded-joints without stress relief.

6. It is an effective method to evaluate and compare the properties of welded-joints by using probability theory and statistical method to treat the fracture toughness of inhomogeneous welded-joints.

#### ACKNOWLEDGEMENTS

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