

The Conditions Low Cycle Crack Initiation and Propagation in Stress Concentration Zones

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SUMMARY The low cycle fracture initiation in the stress concentration zones is determined by cyclic and one-side plastic strain accumulation; the crack propagation is determined by elastoplastic strain intensity. The calculated elastoplastic strain fracture kinetics are experimentally confirmed.

INTRODUCTION The number of cycles N_f needed for fracture in the stress concentration zones under low cycle loading is equal to the sum of numbers of cycles till crack initiation N_c and for crack propagation N_p ($N_f = N_c + N_p$). The value N_c is determined [1] taking into account the cycle-by-cycle changes of plastic strains on the base of linear addition of quasistatic d_s and fatigue d_f damages

$$d_s + d_f = 1 \quad (1)$$

The value d_s is equal to the ratio of one-side accumulated strain $\bar{\epsilon}_{pmax}^{(k)}$ to the corresponding critical value under simple tension e_f , the value d_f denoting the linearly accumulated fatigue damage under cyclic strain of variable amplitude $\bar{\epsilon}_{pamax}^{(k)}$ being determined on the base of power type fatigue curve equation by Manson-Coffin. The values of stress $\bar{\sigma}$ and strain $\bar{\epsilon}$ denote the ratio of the respective values to the values of the yield point $\bar{\sigma}_y$ and e_y ($\bar{\sigma} = \sigma/\bar{\sigma}_y$, $\bar{\epsilon} = e/e_y$). The value N_p is found by integrating the expression for the crack propagation rate $d\ell/dN$ depending on the level of nominal stress $\bar{\sigma}_n$, on the elastic stress concentration factor α_σ and on cyclic properties of the material. The strain criteria of fracture are used for evaluation of N_c and N_p .

RESULTS The maximum strain values $\bar{\epsilon}_{max}^{(o)}$ in the stress concentration zones in the zero half-cycle are equal:

$$\bar{\epsilon}_{max}^{(o)} = \bar{\epsilon}_n \cdot K_e \quad (2)$$

where $\bar{\epsilon}_n$ is the nominal strain; K_e is the strain concentration factor. The stress-strain diagram is approximated by the power equations:

$$\bar{\sigma}_n = \bar{\epsilon}_n^{m_0} \quad \text{and} \quad \bar{\sigma}_{max}^{(o)} = \bar{\epsilon}_{max}^{(o)m_0} \quad (3)$$

where m_0 is the hardening factor in the elasto-plastic region. The value K_e in elasto-plastic condition is calculated /2/ on the base of the following modifications of the Neuber formula:

$$K_e = \frac{\alpha_\sigma^{2/(1+m_0)} \cdot \bar{\sigma}_n^{-(1+m_0)(1+m_0)}}{(\alpha_\sigma \cdot \bar{\sigma}_n)^{n(1+m_0)[1-(\bar{\sigma}_n-1/\alpha_\sigma)]/(1+m_0)}} \quad \text{for } \bar{\sigma}_n \leq 1 \quad (4)$$

$$K_e = \frac{\alpha_\sigma^{2/(1+m_0)}}{(\alpha_\sigma \cdot \bar{\sigma}_n)^{n(1+m_0)[1-(\bar{\sigma}_n-1/\alpha_\sigma)]/(1+m_0)}} \quad \text{for } \bar{\sigma}_n \geq 1$$

in which $n = 0.5$. The maximum values of stresses $\bar{\sigma}_{max}^{(k)}$ and strains $\bar{\epsilon}_{max}^{(k)}$ in the stress concentration zones at half-cycle k are determined /1/ from the stress-strain diagram according to (3): $\bar{\sigma}_{max}^{(k)} = \bar{\epsilon}_{max}^{(k)m(k)}$ ($m(k)$ is the cyclic hardening factor). The values $\bar{\sigma}_{max}^{(k)} = \bar{\sigma}_{max}^{(k)}/S'_y$ and $\bar{\epsilon}_{max}^{(k)} = \bar{\epsilon}_{max}^{(k)}/\bar{\epsilon}_y$ are measured in coordinates with a centre at the point of unloading. The value $m(k)$ found from experimental data for $S'_y = 2\bar{\sigma}_y$ and $\bar{\epsilon}_y = 2\bar{\epsilon}_y$ is:

$$m(k) = \lg \bar{\epsilon}_{max}^{(o)m_0} / \lg \left[\bar{\epsilon}_{max}^{(o)m_0} + \frac{A}{2} (\bar{\epsilon}_{max}^{(o)} - 1) F(k) \right] \quad (5)$$

where A is the material characteristic. The function $F(k)$ for cyclic stable, hardening and softening materials are /1/: $F(k) = 1$; $F(k) = (k-1)^{-B(\bar{\epsilon}_{max}^{(o)}-1)}$; $F(k) = \exp C(\bar{\epsilon}_{max}^{(o)}-1)$ respectively (B and C are parameters of the cyclic stress-strain diagram). Then $\bar{\epsilon}_{max}^{(k)} = \bar{\epsilon}_n \cdot K_e^{(k)}$ and $\bar{\epsilon}_{apmax}^{(k)} = \frac{1}{2} (\bar{\epsilon}_{max}^{(k)} + 2) = \frac{\bar{\epsilon}_{max}^{(k)}}{2}$

The strain concentration factor $K_e^{(k)}$ is evaluated by (4) substituting $m(k)$ for m_0 . The one-side accumulated strain $\bar{\Delta}_{max}^{(k)}$ during the half-cycle k is equal to the difference of maximum strain for half-cycles k and $k-1$ ($\bar{\Delta}_{max}^{(k)} = \bar{\epsilon}_{max}^{(k)} - \bar{\epsilon}_{max}^{(k-1)}$). The total strain $\bar{\epsilon}_{pmax}^{(k)} \cong 2\bar{\epsilon}_{pmax}^{(k)} = 2 \int \bar{\Delta}_{max}^{(k)} dk$.

The damages d_s and d_p in accordance with /1/ are as follows: $d_s = \bar{\epsilon}_{pmax}^{(k)} / \bar{\epsilon}_p = \frac{2}{\bar{\epsilon}_p} \int_0^k \bar{\Delta}_{max}^{(k)} dk$; $d_p = \frac{1}{2(\bar{\epsilon}_p/4)^{1/m}} \int_0^k \bar{\epsilon}_{apmax}^{(k)/m} dk$ (6)

where m is the power index of the Manson-Coffin equation ($m \cong 0.5$) and $\bar{\epsilon}_p = \frac{1}{\bar{\epsilon}_y} \ln \frac{1}{1-\psi}$ (ψ is the reduction in area).

The number of half-cycles $k_c = 2N_c$ till the crack initiation is determined /1/ by (1) as the upper limit of integrals (6).

The following cyclic loading is accompanied by crack propagation and a variation of the strain at the crack tip. The evaluation of local strains at the crack tip is carried

by using (4) and value of $\alpha_\sigma = \bar{K}_I / (\sqrt{2\pi\tau}) \bar{\sigma}_n$ (7) where \bar{K}_I is the stress intensity factor; τ is distance from

crack tip. Neglecting $1/\alpha_\sigma$ for small τ in (7), we obtain: $K_e = \frac{1}{\bar{\sigma}_n} (\bar{K}_I / \sqrt{2\pi\tau})^{[2-n(1+m_0)(1-\bar{\sigma}_n)]/(1+m_0)}$ for $\bar{\sigma}_n \leq 1$

$$K_e = \frac{1}{\bar{\sigma}_n^{2/(m_0+1)}} (\bar{K}_I / \sqrt{2\pi\tau})^{[2-n(1+m_0)(1-\bar{\sigma}_n)]/(1+m_0)} \quad \text{for } \bar{\sigma}_n \geq 1 \quad (8)$$

The strain near the crack tip can be determined using (2): $\bar{\epsilon}_{max}^{(o)} = \bar{\epsilon}_n \cdot K_e = \bar{K}_{Ie} (2\pi\tau)^{p_{Ie}/2}$ where \bar{K}_{Ie} is the strain intensity factor; p_{Ie} -index

of power. From (8) and (9) it follows that $\bar{K}_{Ie} = \bar{K}_I^{p_{ke}}$ for $\bar{\sigma}_n \leq 1$; $\bar{K}_{Ie} = \bar{\sigma}_n^{(1-m)/(1+m)}$ for $\bar{\sigma}_n \geq 1$ (10)

$$p_{ke} = [2-n(1+m_0)(1-\bar{\sigma}_n)]; \quad p_{Ie} = p_{ke}/2 \quad (11)$$

The crack propagation condition is assumed as the reaching of local strain $\bar{\epsilon}_{max}^{(k)}$ at the crack tip to a critical value $\bar{\epsilon}_{fc}$. The fracturing zone length τ_f is obtained from (9) for $\bar{\epsilon}_{max}^{(k)} = \bar{\epsilon}_{fc}$, taking into account the increasing of the initial crack length with the nominal stress to $\bar{\sigma}_n$

(by the cut and try method): $\tau_f = \frac{1}{2\pi} (\bar{K}_{Ie} / \bar{\epsilon}_{fc})^2 [1 - \frac{1}{2\pi\bar{\sigma}_n} (\bar{K}_{Ie} / \bar{\epsilon}_{fc})^2]^{-1}$ (12)

The crack propagation rate can be calculated from (12) in the case of cyclic loading with a strain intensity factor

range $\Delta \bar{K}_{Ie}$ (in coordinate " $\bar{\sigma} - \bar{\epsilon}$ " for $d\bar{\epsilon}/dN = \tau_f$ and $m_0 = m(k)$) $d\bar{\epsilon}/dN = \frac{1}{2\pi} (\Delta \bar{K}_{Ie} / \bar{\epsilon}_{fc})^2 [1 - \frac{1}{2\pi\bar{\sigma}_n} (\Delta \bar{K}_{Ie} / \bar{\epsilon}_{fc})^2]^{-1}$ (13)

If the crack length is subcritical $\frac{1}{2\pi\bar{\rho}_0}(\Delta\bar{K}_{Ic}/\bar{\epsilon}_{fc})^2 \ll 1$
 $\frac{d\ell}{dN} = \frac{1}{(1/2\pi\bar{\epsilon}_{fc})^2 \Delta\bar{K}_{Ic}^2} \quad (14)$
 The value $d\ell/dN$ from (9) and (14) is proportional to the square of the range of elasto-plastic low-cycle strain at the crack tip $(d\ell/dN \sim (2\bar{\epsilon}_{amax}^{(k)})^2)$. Then, from (10), (13) and (14):
 $\frac{d\ell}{dN} = C_{15} \Delta\bar{K}_{Ic}^2 [1 - \frac{C_{16}}{\bar{\epsilon}_{fc}^2} (\Delta\bar{K}_{Ic})^2]^{-1} \quad (15)$
 where $\gamma = 2\rho_{ke}$ and $C_{15} = 1/2\pi\bar{\epsilon}_{fc}^2$. For the values $m(k) = 0.02$ to 0.3 and $\bar{\rho}_n = 0.3$ to 1 the value $\gamma = 2.7$ to 3.93 , this being in agreement with results of [3,4,5].

In this connection cyclic softening Cr-Mo-V and low-carbon (0.21% C) cyclic stable steels were tested under constant loading amplitudes for different asymmetry factors τ_σ within -1.2 and $+0.6$ and values α_σ . The strain measurements were made by the grid method with a grid pitch of 0.1mm . The interrelation between the values $d\ell/dN$ and $2\bar{\epsilon}_{amax}^{(k)}$ independent on loading conditions is shown in Fig.1. The experimental data are in agreement with (9) and (14). A comparison of calculated data and experimental results at the stages of crack initiation ($N \leq N_c$) and crack propagation ($N_c \leq N \leq N_f$) are shown in Fig.2 and Fig.3. The final fracture is determined by reaching at the crack tip. $\bar{\epsilon}_{max}^{(k)} = \bar{\epsilon}_{fc}$. Critical value of $\bar{\epsilon}_{fc}$ remains constant for different concentration levels and fracturing numbers of cycles.

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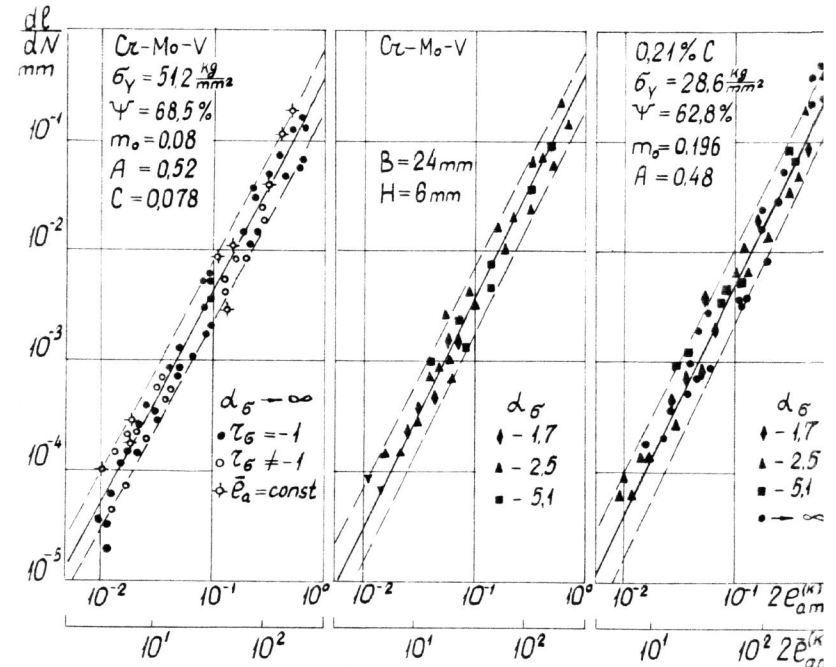


Fig.1 Crack propagation rate versus local strain range

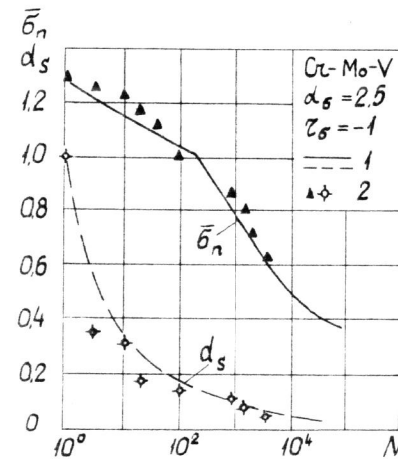


Fig.2 Characteristics of crack initiation (1 - analytical, 2 - experimental).

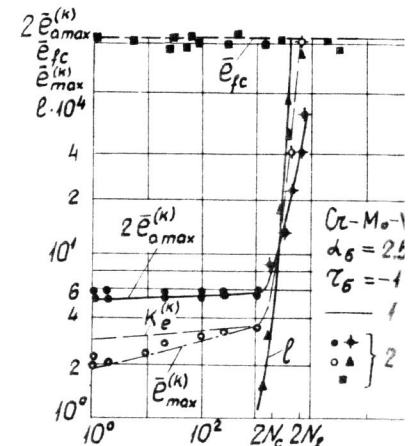


Fig.3 Conditions of crack initiation and propagation (1 - analytical, 2 - experimental).