

A Quantitative Model for the Temperature, Strain Rate, and Grain Size Dependence of Fracture Toughness in Low Alloy Steel

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Prior investigations have established that cleavage fracture initiates in cracked specimens of low alloy steel when the maximum tensile stress level in the plastic zone ahead of the crack, σ_{yy}^{\max} , reaches a critical value σ_f^* .⁽¹⁾ This value is known as the microscopic cleavage strength. To build the local stress level up to σ_f^* from the unconstrained value of the tensile yield strength, σ_y , we have proposed⁽¹⁾ that the plastic zone must reach a critical value R_c that is given by

$$R_c = \rho [\exp(\sigma_f^*/\sigma_y - 1) - 1] \quad (1)$$

where ρ is the root radius of the crack.

The plane strain fracture toughness, $K_{Ic}(\rho)$ is that value of stress intensity K at which $R = R_c$.

From sharp crack fracture mechanics, the plane strain plastic zone size is given by⁽²⁾

$$R \cong 0.12 \left(\frac{K_I}{\sigma_y} \right)^2$$

so that

$$R_c = 0.12 \left(\frac{K_{Ic}}{\sigma_y} \right)^2 \quad (2)$$

and consequently, for a crack of root radius ρ , equations (1) and (2) give

$$K_{Ic}(\rho) = 2.9 \sigma_y [\exp(\sigma_f^*/\sigma_y - 1) - 1]^{1/2} \sqrt{\rho} \quad (3)$$

Figure 1 indicates that the fracture toughness of carbon/manganese steel measured in three point bending at -196°C does indeed vary with $\sqrt{\rho}$, for $\rho > \rho_0$. For $\rho < \rho_0$ the fracture toughness is the same as that

evaluated by standard ASTM specimens which have been notched and fatigue cracked ($\rho=0$) prior to testing. Consequently,

$$K_{IC} = K_{IC}(\rho_0) = 2.9 \sigma_y [\exp(\sigma_f^*/\sigma_y - 1) - 1]^{1/2} \sqrt{\rho_0} \quad (4)$$

The parameter ρ_0 is known as the "effective root radius" of the crack. (3) It is a measure of the extent of the process zone ahead of the crack over which the critical stress σ_f^* must exist to initiate cleavage. It has been proposed (4) that ρ_0 is determined by some microstructural feature, such as grain size or spacing of inclusions or slip or twin bands. If σ_f^* and ρ_0 are independent of temperature, then it is possible to determine the temperature dependence of K_{IC} and also the dynamic fracture toughness K_{ID} from the temperature and strain rate dependence of σ_y .

Equation (4) predicts that the plane strain cleavage fracture toughness depends only on three parameters, a) σ_f^* , the microscopic cleavage strength, b) σ_y , the yield strength, and c) ρ_0 , the effective limiting crack sharpness.

The present research was designed to determine the effect of microstructure and composition on σ_f^* , σ_y , and ρ_0 , and therefore determine a direct expression for K_{IC} in terms of microstructural variables. Small specimens of Charpy dimension (10 x 10 x 60mm.) with 2mm. deep notches of variable radius were used to evaluate $K_{IC}(\rho)$ and $K_{IC} = K_{IC}(\rho_0)$ at -196°C . σ_y was measured separately at the strain rate which exists below the notch during bending. These parameters were then used to determine ρ_0 and σ_f^* from Equation (4).

A low carbon/manganese steel was used for initial investigation. After machining, the specimens were heat-treated to obtain three different grain sizes. This was done by varying the austenitizing temperature and the cooling rate. Three grain sizes $d=1.3 \times 10^{-3}$ in., 2.5×10^{-3} in., and 3.5×10^{-3} in., were obtained. All specimens then were annealed at 1000°F for three hours to assure the same substructure for the different grain sizes.

The specimens were broken in slow bending at -196°C , and for each case values of K_{IC} and ρ_0 were determined from Figure 1. The yield strength σ_y was measured separately in a tensile test and Equation (4)

was then used to determine σ_f^* . The experimental results on the specimens of varying grain size indicate that:

$$\begin{aligned} \sigma_y &= 44 + 2.7d^{-1/2} & (\dot{\epsilon} = 8.3 \times 10^{-3} \text{ sec}^{-1}) \\ \sigma_f^* &= 26 + 6.4d^{-1/2} & d^{-1/2} > 11.0 \\ \rho_0 &\cong 3d \end{aligned} \quad (5)$$

with σ_y and σ_f^* in units of ksi and d in inches.

Substituting into Equation (4) we get:

$$K_{IC} = (220d^{1/2} + 13.5) \left[\exp\left(\frac{3.7d^{-1/2} - 18}{2.7d^{-1/2} + 44}\right) - 1 \right]^{1/2} \quad (6)$$

An instrumented Charpy impact machine was used to duplicate the above results for the dynamic case. Figure 1 shows the results of this work. Dynamic yield strength was determined by extrapolating the general yield load, P_{GY} , measured at higher temperatures down to -196°C and then using the Green and Hundy (5) relation $\sigma_y = 33.3P_{GY}$. It was found that:

$$\begin{aligned} \sigma_y &= 49 + 3.8d^{-1/2} & (\dot{\epsilon} = 100 \text{ sec}^{-1}) \\ \sigma_f^* &= 26 + 6.4d^{-1/2} \\ \rho_0 &\cong 3d \end{aligned} \quad (7)$$

And substituting in Equation (4) we get:

$$K_{ID} = (245d^{1/2} + 19) \left[\exp\left(\frac{2.6d^{-1/2} - 24}{3.8d^{-1/2} + 45}\right) - 1 \right]^{1/2} \quad (8)$$

As shown in Figure 2, the experimental values of K_{IC} and K_{ID} are in excellent agreement with these relations. Equation (6) and (8) predict that for this particular steel subsequent refinement in grain size produces only a slight increase in K_{IC} and K_{ID} at -196°C . This fact, which has also been observed for mild steel, arises because although the microscopic cleavage strength σ_f^* is increased by grain refinement, the effective crack tip radius ρ_0 is reduced. The decrease in ρ_0 with

decreasing grain size is due to the fact that the stress level should reach σ_f^* over a few grains below the notch for cleavage to occur. This distance and therefore, ρ_0 , decrease as the grain size is decreased.

Equation (5) and (7) indicate that strain rate has no effect on σ_f^* or ρ_0 which indicates that σ_f^* and ρ_0 are probably independent of temperature. The decrease in K_{Ic} with increasing strain rate simply results from increasing σ_y with increasing strain rate.

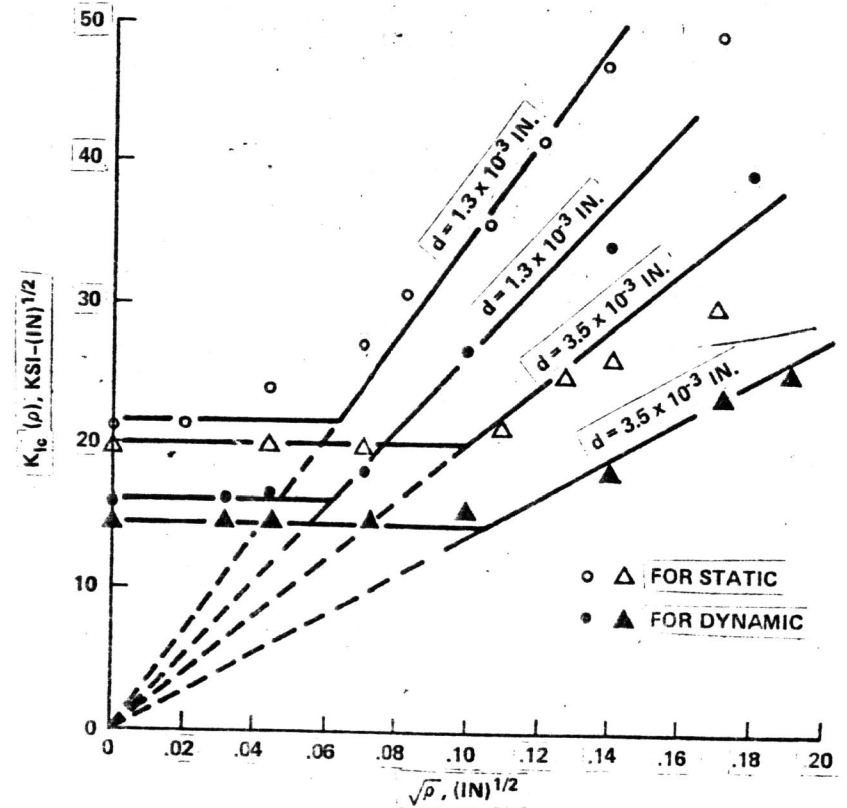


Figure 1. Variation of $K_{Ic}(\rho)$ with Notch Root Radius at -196°C

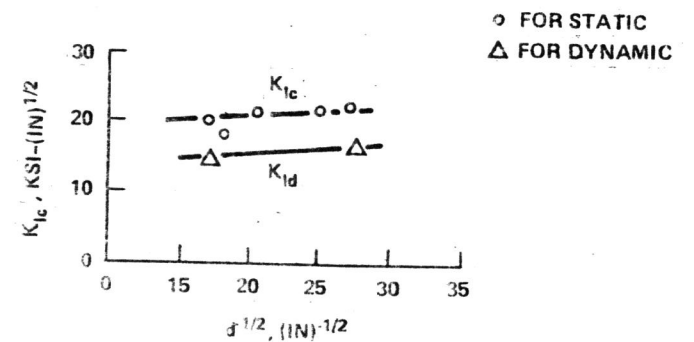


Figure 2. Variation of Fracture Toughness with Grain Size at -196°C

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References

1. T.R. Wilshaw, and C.A. Rau, Jr., A.S. Tetelman, The International J. of Fracture Mechanics, Vol. 4, No. 2 (June 1968) pp. 147-157.
2. F.A. McClintock and G.R. Irwin, Fracture Toughness Testing, ASTM STP 381, (1965) pp. 84.
3. J. Malkin and A.S. Tetelman, Engr. Fracture Mechanics, Vol. 3, (1971) pp. 151-167.
4. A.H. Cottrell, The Mechanical Properties of Matter, John Wiley & Sons (1964) pp. 365.
5. A.P. Green and B.B. Hundy, J. Mech. Phys. Solids, 4, (1956) pp. 128.
6. S. Ensha, "The Effect of Microstructure on the Fracture Toughness of Low Alloy Steel." Doctoral thesis, Materials Department, University of California, Los Angeles.